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'' (g2/F51Tf16.34070TD(B)Tj/F41Tf0.65890TD(2)Tj/F21Tf0.49720TD[()-385.8(has)-387.1(its)]TJ/F41Tf

Li, the *hR1* structure transforms to the *cI16* structure at about 35 GPa, then at about 75 GPa it transforms to the *C2* structure, and then at about 98 GPa it transforms to the *Aba2* structure, which persists up to at least 160 GPa; for Cs, the *Cmca* structure transforms to the *dhcp* structure at about 60 GPa, which persists up to at least 200 GPa. The transition pressures are in agreement with the experimental results [3–5]. Random search techniques [24] and the evolutionary algorithm method [25,26] were previously applied to search the ground states of elemental lithium by Richard et al. [6], Yasuda et al. [7], and respectively by Gao et al. [8]. (Experimental)

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$$\Delta H_{\text{VD}} = \frac{1}{4} (H_{\text{LiCs}} - H_{\text{Li}} - \frac{1}{2} H_{\text{Cs}}) + P(V_2 - V_1)/2 \quad (1)$$

The charge-transfer energy ΔH_{CT} is the enthalpy change when combining bcc Cs and bcc Li at the constant-volume V_m to form LiCs (B2),

$$\Delta H_{\text{CT}} = H_{\text{LiCs}} - H_{\text{Li}} - H_{\text{Cs}} \quad (2)$$

ΔH_{VD} and ΔH_{CT} are shown as the red line and the green line in Fig. 4(a), respectively. We see that ΔH_{VD} is positive and increases rapidly as pressure increases, the main effect coming from the $P(V_2 - V_1)/2$ term (dashed line). At zero pressure, ΔH_{VD} (520 meV/atom) is large enough to make the Li-Cs system phase separating. In contrast, ΔH_{CT} is negative and decreases rapidly as pressure increases. At 160 GPa, $\Delta H = -258$ meV/atom and ΔH_{CT} is large enough to induce ordering in Li-Cs.

The electronic structures of LiCs (B2) are shown in Figs. 4(b) and 4(c). At zero pressure the dispersions of the $\text{Li } d_{1s}$, $\text{Cs } d_{5s}$, and $\text{Cs } d_{5p}$ bands are all small. At 160 GPa, the $\text{Li } d_{1s}$ band still has negligible dispersion, while the dispersion of the $\text{Cs } d_{5s}$

high-pressure compounds between two elements in the same column of the periodic table (Li-Cs) is perhaps reminiscent of compound formation and charge transfer between Ag-K [27] or Li-Be [17], although the latter two cases are perhaps less surprising than Li-Cs as they represent a far larger chemical disparity (hence, reactivity) between the constituent elements that belong to different columns in the periodic table.

Li_7Cs ($Cmmm$)^x. . . 1D- . . . The electronic structure of Li_7Cs ($Cmmm$) at 80 GPa exhibits interesting features of 1D-like bands. Its density of states at the Cs $5s$ and Cs $5p$ bands [see Fig. 6(a)] displays a remarkable feature of typical 1D electronic structure, the emergence of which in a three-dimensional environment is rather unexpected. Figure 6(b) shows the band structure of Li_7Cs ($Cmmm$) at 80 GPa. The Cs $5s$ and Cs $5p$ bands display strong dispersion only along the Γ - X_3 direction, but are nearly dispersionless along other directions. The 1D band structure is caused by the unique crystal structure of Li_7Cs ($Cmmm$) [see the inset of Fig. 3(a)]. The Cs chains are embedded in the three-dimensional Li networks and are separated from each other. Along the chain direction, the Cs atoms are close to each other and their cores overlap.